

Article

Defect-Engineered Carbon-Spinel Interfaces for Enhanced Periodate Activation for Bisphenol A Degradation

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Abstract

Developing efficient and sustainable catalysts for advanced oxidation processes (AOPs) to remove endocrine-disrupting compounds remains a critical challenge. In this study, a defect-engineered $\text{MnFe}_2\text{O}_4@\text{SBC}$ composite was synthesized by loading spinel MnFe_2O_4 onto sewage sludge-derived biochar (SBC) prepared at different calcination temperatures, and applied for efficient periodate (PI) activation toward bisphenol A (BPA) degradation. The catalytic performance exhibited a volcano-type dependence on calcination temperature, with $\text{MnFe}_2\text{O}_4@\text{SBC-750}$ achieving the highest BPA removal efficiency (98.6% within 30 min). Structural characterization revealed that $\text{MnFe}_2\text{O}_4@\text{SBC-750}$ possessed an optimized carbon structure with a balance between defect sites and graphitized domains. Mechanistic investigations demonstrated that multiple reactive oxygen species, including $\cdot\text{OH}$, $\text{O}_2^{\cdot-}$, $\text{IO}_3^{\cdot}$ and $^1\text{O}_2$, were involved in BPA degradation. LC-MS analysis identified key transformation intermediates and proposed degradation pathways, while toxicity assessment confirmed reduced ecological risks after treatment. Density functional theory (DFT) calculations indicated that $\text{MnFe}_2\text{O}_4@\text{SBC}$ significantly enhanced PI adsorption and activation by promoting interfacial electron transfer and elongating the I-O bond in IO_4^- . Notably, $\text{MnFe}_2\text{O}_4@\text{SBC-750}$ exhibited the strongest electron transfer capability, attributed to the optimal regulation of defect density and graphitization degree, which facilitated π -d electronic coupling at the MnFe_2O_4 -SBC interface. Overall, this work elucidates the critical role of defect regulation in spinel biochar-based catalysts for oxidant activation and provides a sustainable strategy for converting sewage sludge into high-performance catalysts for water purification.

Keywords: $\text{MnFe}_2\text{O}_4@\text{SBC}$; periodate activation; advanced oxidation technology; defect engineering; π -d electronic coupling



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1. Introduction

Bisphenol A (BPA) is a widely used industrial chemical that plays a significant role in the production of polycarbonate plastics and epoxy resins [1]. These materials are critical in the manufacturing of everyday products such as beverage containers, protective coatings and food packaging. Due to its chemical stability, malleability and heat resistance, BPA has been utilized extensively across various industries [2]. However, BPA is an endocrine-disrupting chemical (EDC) that poses serious health risks by interfering with hormone

regulation and disrupting endocrine functions, which leads to various health issues such as developmental and metabolic disorders [3]. The environmental impact of BPA is equally concerning, as it is highly persistent and resistant to degradation, often accumulating in water sources due to wastewater discharge from industrial processes and leaching from landfills [4]. Groundwater contamination by BPA has reached alarming levels in some regions, with concentrations exceeding $740 \mu\text{g L}^{-1}$ in certain areas affected by landfill leachates [5]. Studies have shown that even low concentrations of BPA can adversely affect aquatic organisms, including fish and invertebrates, causing developmental abnormalities and reproductive disruptions [6], as well as BPA ubiquity in human biological samples, such as urine, blood and breast milk [7]. Given these concerns, the development of advanced, cost-effective, and environmentally friendly treatment methods for BPA degradation is critical to mitigating its impact on both human health and the environment.

Advanced oxidation processes have become a promising green and environmentally friendly new technology, with periodate (PI)-based AOPs attracting increasing attention in recent years [8]. PI is a high-valent iodine oxidant featuring a redox potential of +1.60 V, excellent stability and low transportation and storage risks compared with conventional oxidants [9]. PI can generate a suite of highly reactive oxygen species (ROS), including iodyl radicals ($\text{IO}_3^\bullet$), hydroxyl radicals ($^\bullet\text{OH}$), singlet oxygen ($^1\text{O}_2$) and superoxide radicals ($\text{O}_2^{\bullet-}$), via I-O bond cleavage, enabling rapid transformation or mineralization of refractory organic pollutants [10]. However, the reaction kinetics of unactivated PI with most organic pollutants remain slow, highlighting the urgent need for efficient activation strategies [11]. Multiple activation pathways have been developed, including transition-metal catalysis, photoactivation, ultrasound, alkali activation, redox-mediated activation and heterogeneous catalysis [12]. Among these, transition-metal-based activation stands out due to its high catalytic efficiency, low energy input and potential for generating multiple ROS synergistically. Particularly, Fe- and Mn-based oxides catalysts have demonstrated remarkable performance in PI activation [13]. Xu et al. [14] demonstrated that MnFe_2O_4 could efficiently activate PI to generate reactive species, such as $^1\text{O}_2$, $\text{IO}_3^\bullet$ and $\text{O}_2^{\bullet-}$, leading to almost 100% removal of ciprofloxacin within 90 min.

Sewage sludge, a by-product of municipal wastewater treatment, is generated in tremendous quantities worldwide and contains a complex mixture of organic matter, heavy metals and pathogenic microorganisms, posing severe environmental risks if improperly disposed of [15]. Converting sewage sludge into biochar (SBC) through pyrolysis presents a sustainable and carbon-negative strategy for sludge resource utilization [16]. SBC exhibits a carbon-rich structure with abundant oxygen-containing surface functional groups, porous architecture and a considerable specific surface area, making it a versatile catalytic substrate for advanced oxidation processes [17]. Numerous studies have demonstrated that SBC can activate PI through mechanisms involving surface functional groups ($-\text{OH}$, $-\text{COOH}$, $-\text{OOH}$), electron-transfer mediation, defect-induced redox reactions and inherent transition-metal species originating from wastewater sludge [18,19]. However, pristine SBC often suffers from limited catalytic efficiency, insufficient redox activity, and structural instability [20]. MnFe_2O_4 spinel has been proven to have excellent activation ability for PI, with high reactivity, ease of recovery and environmental and economic friendliness [21]. However, nanoscale MnFe_2O_4 tends to aggregate and reduces the contact between surface-active sites and PI, which decreases its catalytic performance. Previous studies have shown that the combination of MnFe_2O_4 with biochar as a catalyst exhibits strong synergistic effects, which not only prevent the aggregation of spinel nanoparticles but also promote the generation of reactive oxygen species and enhance catalytic stability [22]. Notably, defect structures in biochar can disrupt the symmetry of the π -conjugated carbon framework and introduce unpaired electrons, thereby inducing localized charge polarization and

enhancing interfacial electron transfer [23]. For instance, Luo et al. [24] demonstrated that neighboring carbon vacancy defects could substantially reconstruct the local electronic structure, giving rise to intensified interfacial charge coupling and thereby promoting the activation of reactant molecules. However, the defect density of biochar is dependent on thermal treatment conditions, and biochar annealed at different temperatures exhibit distinct defect types and distributions [25]. These differences can directly affect the extent of π -electron delocalization, the strength of interfacial electronic coupling, and overall charge transport behavior [26]. Although previous studies have revealed the positive role of defect structures in catalytic reactions, how the regulation of defect density influences the electronic structure of the carbon support–metal interface and the overall catalytic performance of the system remains insufficiently understood. To our knowledge, no prior studies have comprehensively elucidated the impact of defect density modulation on catalyst-wide electronic interactions and reactivity.

Therefore, in this study, municipal sewage sludge was employed as the precursor to prepare a series of defect-engineered SBC via pyrolysis at different temperatures, which were further loaded with MnFe_2O_4 to construct $\text{MnFe}_2\text{O}_4@\text{SBC}$ composite catalysts. The removal efficiency and degradation mechanism of BPA were systematically investigated through structural characterization, reactive species identification, and kinetic analyses. In combination with density functional theory (DFT) calculations, the regulatory effects of SBC defect density and graphitization degree at different calcination temperatures on the interfacial electronic structure of $\text{MnFe}_2\text{O}_4@\text{SBC}$, as well as on PI adsorption and activation behavior, were elucidated. The results demonstrate that the synergistic interplay between defect density and π -d electronic coupling plays a decisive role in determining the catalytic activity. This work provides important theoretical insights into the defect engineering of carbon-based composite catalysts for enhanced catalytic performance.

2. Materials and Methods

2.1. Materials and Synthesis of Catalysts

Details of these chemicals and reagents can be found in Supplementary Materials Text S1. The raw sewage sludge was collected from the Hedong Sludge Treatment Plant (Beijing, China). Typically, the sludge was first cleaned three times with deionized water (DI water) in an ultrasonic cleaning machine. After natural sedimentation, the sludge was dried in an oil bath at 105 °C until a constant weight was achieved. It was then ground and sieved through a 60-mesh screen to obtain the biochar precursor. The precursor was then pyrolyzed in a tube furnace at a specified temperature with a heating rate of 5 °C min^{−1} for 2 h, under a nitrogen flow of 300 mL L^{−1}. After natural cooling, the biochar produced from the pyrolysis was washed repeatedly with ethanol and DI water, and then dried at 60 °C for 12 h to obtain SBC. The $\text{MnFe}_2\text{O}_4@\text{SBC}$ catalyst was further synthesized by hydrothermal treatment. First, 0.095 g of Manganese chloride tetrahydrate ($\text{MnCl}_2 \cdot 4\text{H}_2\text{O}$, 12 mM), 0.399 g of Ferric nitrate nonahydrate ($\text{Fe}(\text{NO}_3)_3 \cdot 9\text{H}_2\text{O}$, 24 mM) (molar ratio of Mn:Fe = 1:2) and 0.6 g of SBC were dissolved in 40 mL of DI water and stirred magnetically for 30 min. The resulting mixture was aged at room temperature for 4 h, and then transferred to a Teflon-lined stainless-steel autoclave for hydrothermal treatment at 180 °C for 12 h. After cooling to room temperature, the product was carefully washed with deionized water and dried in an oven at 60 °C to obtain $\text{MnFe}_2\text{O}_4@\text{SBC}$. Additionally, MnFe_2O_4 was prepared following the same procedure but without adding SBC.

2.2. Experimental Procedures

Unless otherwise specified, all degradation experiments were conducted in a 100 mL beaker at 25 ± 1 °C with a stirring speed of 350 rpm. First, the BPA stock solution

(100 mg L⁻¹) was diluted to the target concentration with DI water in the beaker and stirred to ensure uniform mixing. Subsequently, a certain amount of MnFe₂O₄@SBC was added to the system to initiate the adsorption process. After equilibrium was reached (30 min), 1 mL of the mixed solution was withdrawn using a syringe and filtered through a 0.22 µm Polytetrafluoroethylene (PTFE) filter membrane. PI stock solution (200 mM) was then immediately added to trigger the catalytic reaction. Samples were taken at specific time points (2, 5, 10, 15, 20 and 30 min), filtered through a 0.22 µm membrane, and immediately treated with 100 µL of 0.1 M sodium thiosulfate (Na₂S₂O₃, 100 mM) solution to remove any remaining PI and reactive species. All experiments were repeated at least three times to obtain the standard deviation. BPA was quantitatively analyzed using a high-performance liquid chromatography (HPLC, Agilent Technologies Inc., Braum, Germany) equipped with a C18 reverse-phase column and UV detector at 278 nm. Additional experimental procedures are provided in Text S2.

2.3. Analytical Methods

The quantification of PI, iodinated toxic by-products and other organic pollutants was performed using HPLC. Detailed HPLC testing methods were provided in Table S1. I₂ and I₃⁻ were identified using the starch colorimetric method. Reactive free radicals were identified by electron paramagnetic resonance (EPR, Bruker Magnetic Resonance, Saarbrücken, Germany), with DMPO used as a spin trap to detect the generated •OH, O₂•⁻ and IO₃• in the MnFe₂O₄@SBC/PI system, and TEMP used as a spin trap for the non-radical ¹O₂. Intermediate products of BPA degradation were analyzed using ultra-performance liquid chromatography–mass spectrometry (UPLC-MS, Thermo Fisher Scientific, Waltham, MA, USA). The toxicity of BPA and its intermediate degradation products was assessed using the Toxicity Estimation Software Tool (T.E.S.T., v5.1.2) by the U.S. Environmental Protection Agency (EPA). Additional characterization methods and instrument parameters are provided in Text S3, and the analytical methods are detailed in Text S4.

2.4. Theoretical Calculations

DFT calculations were performed using the Vienna ab initio simulation package (VASP.5.4.4) and Gaussian 16. The detailed computational procedures and parameters were provided in Text S5.

3. Results and Discussions

3.1. Catalyst Characterization

The surface morphology of SBC, MnFe₂O₄ and MnFe₂O₄@SBC was investigated by scanning electron microscope (SEM). As shown in Figure 1a,d, SBC exhibited a relatively flat surface with occasional block-like and sheet-like regions. EDS analysis revealed that SBC contained abundant C and O, along with Mn and Fe (Figure S1a). However, these Mn and Fe containing species originated from the complex metal matrix inherent to municipal sludge and were transformed into poorly soluble oxides or silicate-bound phases during pyrolysis at 550 °C. Such thermally stabilized forms did not dissolve or participate in the subsequent hydrothermal synthesis. The synthesized MnFe₂O₄ sample displayed nano-sized granular particles with aggregation, and its EDS spectrum confirmed the presence of O, Mn and Fe (Figures 1b,e and S1b). The SEM images of MnFe₂O₄@SBC showed that MnFe₂O₄ nanoparticles and small clusters are attached to the surface of the biochar, indicating that the carbon matrix effectively inhibited the particle agglomeration commonly observed in pure MnFe₂O₄ (Figure 1c,f). Elemental mapping of MnFe₂O₄@SBC (Figure S1c) revealed homogeneous distributions of C, O, Mn and Fe, confirming that MnFe₂O₄ has been successfully loaded onto SBC. Transmission electron microscope (TEM) images further

confirmed the spinel nanostructure of $\text{MnFe}_2\text{O}_4@\text{SBC}$ on the surface (Figure 1g). High-resolution transmission electron microscope (HRTEM) characterization of $\text{MnFe}_2\text{O}_4@\text{SBC}$ displayed lattice fringes with an interplanar spacing of approximately 0.261 nm (Figure 1h), consistent with the (311) plane of spinel MnFe_2O_4 [14]. In addition, selected area electron diffraction (SAED) pattern (Figure 1i) displayed clear diffraction rings corresponding to the (311) lattice plane of MnFe_2O_4 [27].

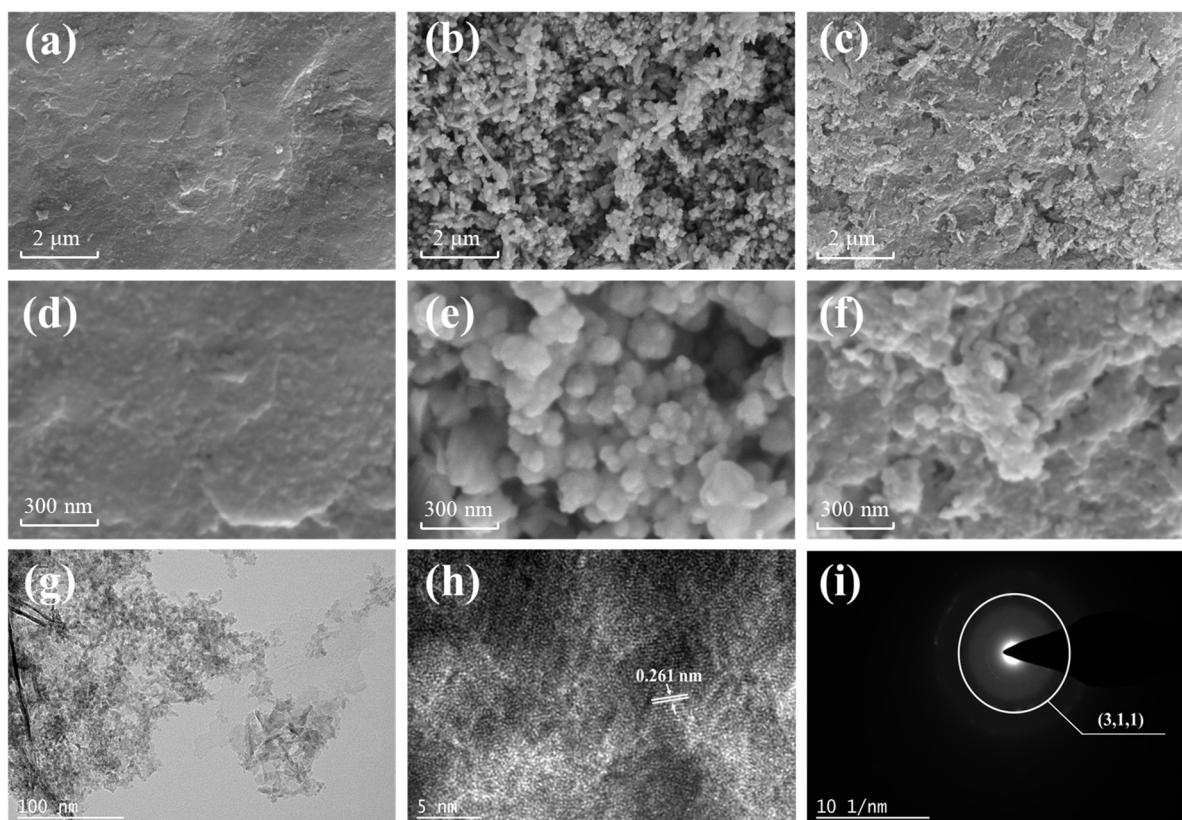


Figure 1. SEM images of 2 μm and 300 nm for (a,d) SBC, (b,e) MnFe_2O_4 and (c,f) $\text{MnFe}_2\text{O}_4@\text{SBC}$, (g) TEM and (h,i) HR-TEM images of $\text{MnFe}_2\text{O}_4@\text{SBC}$.

The surface area and pore structure of $\text{MnFe}_2\text{O}_4@\text{SBC}$ were analyzed using N_2 adsorption–desorption isotherms and pore size distribution analysis. As shown in Figure S2a and Table S2, SBC exhibited a specific surface area of $86.09 \text{ m}^2/\text{g}$, with an average pore diameter of 8.57 nm. The specific surface area of MnFe_2O_4 is $27.50 \text{ m}^2/\text{g}$, with an average pore diameter of 18.20 nm (Figure S2b). The pore size distribution was irregular and variability. In comparison, the $\text{MnFe}_2\text{O}_4@\text{SBC}$ composite exhibited a substantial increase in surface area, reaching $248.70 \text{ m}^2/\text{g}$ and an average pore diameter of 10.76 nm (Figure S2c). This substantial increase in surface area could be attributed to the synergistic effect between MnFe_2O_4 nanoparticles and the biochar matrix, which enhanced the overall porosity and facilitates greater exposure of active sites. The N_2 adsorption–desorption isotherm of the composite shows a type IV adsorption isotherm with a H3 hysteresis loop [28]. Furthermore, the pore size distribution revealed that the $\text{MnFe}_2\text{O}_4@\text{SBC}$ composite exhibits a well-developed pore structure with a predominant pore volume in the 10–20 nm range. The results indicated the presence of micropores and mesopores, which were beneficial for facilitating the transport of pollutants and reactive species.

The X-ray diffraction (XRD) patterns of $\text{MnFe}_2\text{O}_4@\text{SBC}$ were recorded to investigate the crystallographic structure of the composite material. As shown in Figure 2a, the unique characteristic peak of $\text{MnFe}_2\text{O}_4@\text{SBC}$ is assigned to the (002) plane of a hexagonal graphite

structure (JCPDS No. 99-0057), which is due to the low loading of MnFe_2O_4 [29]. The XRD pattern of MnFe_2O_4 presents several prominent diffraction peaks, which are consistent with those of spinel-type MnFe_2O_4 (JCPDS No. 38-0430), confirming the successful synthesis of MnFe_2O_4 [30]. The diffraction peaks observed at $2\theta = 29.6^\circ, 34.8^\circ, 42.3^\circ, 56.0^\circ$ and 61.5° correspond to the (202), (311), (004), (333) and (440) crystallographic planes of MnFe_2O_4 , respectively. The average crystallite size of MnFe_2O_4 calculated from the Scherrer equation using the (311) diffraction peak was approximately 17.6 nm, indicating the formation of nanosized spinel MnFe_2O_4 particles on the SBC surface. The relatively small crystallite size could expose more active sites and facilitate interfacial catalytic reactions during PI activation. Additionally, the XRD pattern showed characteristic peaks at $24.1^\circ, 33.2^\circ, 35.6^\circ, 39.3^\circ, 40.9^\circ, 43.5^\circ, 49.5^\circ$ and 54.1° , which correspond to the (012), (104), (110), (006), (113), (202), (024) and (116) planes of Fe_2O_3 (JCPDS No. 87-1166) [31]. These results indicate that MnFe_2O_4 was successfully loaded onto SBC and that a small amount of Fe_2O_3 was generated during the synthesis process.

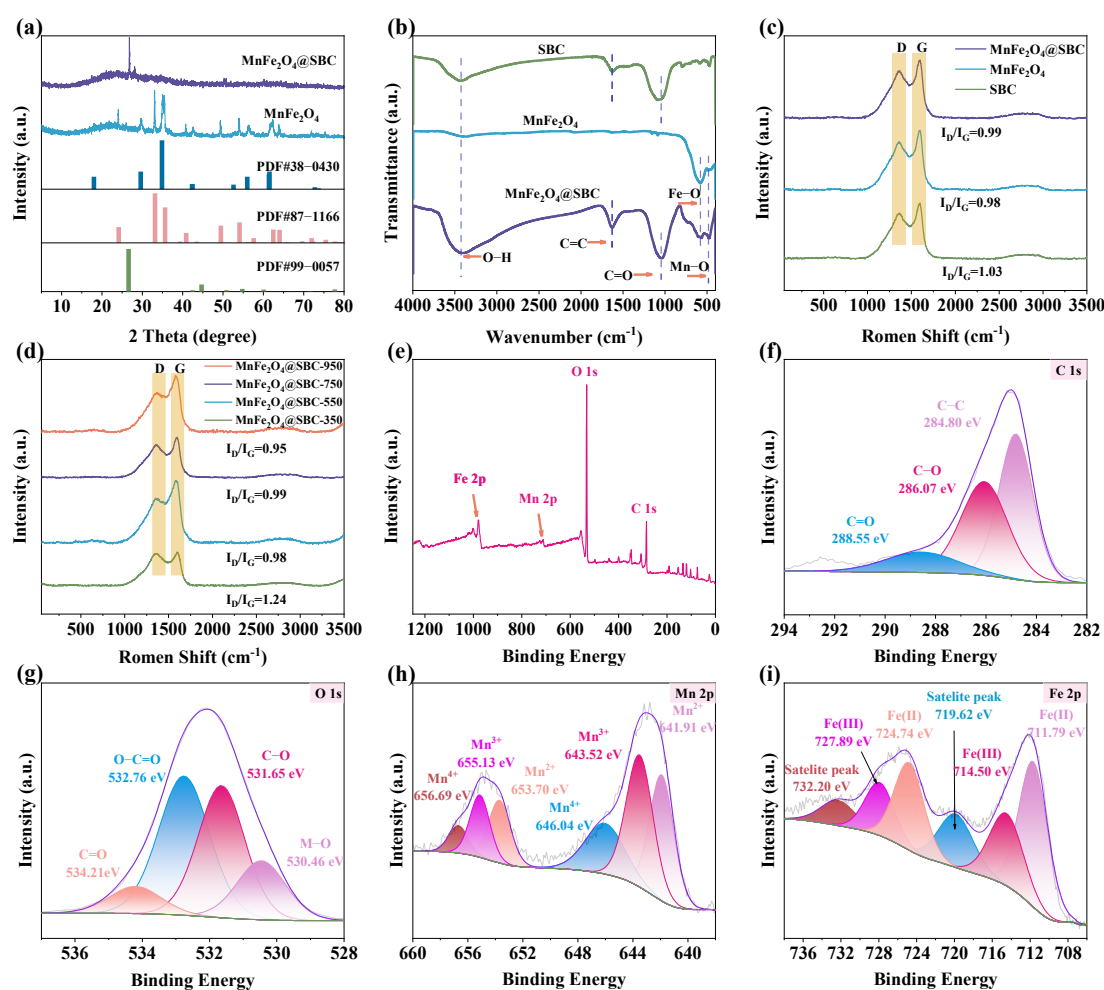


Figure 2. (a) XRD patterns, (b) FT-IR spectra and (c) Raman spectra of SBC, MnFe_2O_4 and $\text{MnFe}_2\text{O}_4/\text{SBC}$, (d) Raman spectra of $\text{MnFe}_2\text{O}_4/\text{SBC}$ synthesized from SBC prepared at different pyrolysis temperatures, (e) XPS spectra of $\text{MnFe}_2\text{O}_4/\text{SBC}$, high-resolution XPS spectra of (f) C1s, (g) O1s (h) Mn2p and (i) Fe2p of $\text{MnFe}_2\text{O}_4/\text{SBC}$.

The Fourier-transform infrared (FTIR) spectra of $\text{MnFe}_2\text{O}_4/\text{SBC}$ are shown in Figure 2b, providing qualitative information on the characteristic peaks of various functional groups. A prominent peak at 3415 cm^{-1} was observed, which is attributed to the O-H stretching vibration of hydroxyl groups, indicating the presence of surface hydroxyl

functionalities on SBC, MnFe_2O_4 and $\text{MnFe}_2\text{O}_4@\text{SBC}$ [32]. At 1627 cm^{-1} , the peak corresponds to the C=C stretching vibration of the graphitic structure within SBC matrix, reflecting the aromatic nature of the material [33]. A peak at 1043 cm^{-1} was observed related to the C=O stretching vibration of carbonyl groups, likely linked to surface oxygen-containing groups [34]. The absorption bands at 584 cm^{-1} and 482 cm^{-1} in MnFe_2O_4 and $\text{MnFe}_2\text{O}_4@\text{SBC}$ are attributed to the stretching vibrations of the Fe-O and Mn/Fe-O bonds, respectively [35]. These peaks confirm the presence of metal-oxide bonds, further supporting the successful incorporation of MnFe_2O_4 nanoparticles into the biochar matrix.

Raman spectroscopy was employed to further investigate the graphitization degree and structural properties of $\text{MnFe}_2\text{O}_4@\text{SBC}$. As shown in Figure 2c, the Raman spectra of SBC, MnFe_2O_4 and $\text{MnFe}_2\text{O}_4@\text{SBC}$ exhibited two prominent peaks. The D band at 1339 cm^{-1} corresponds to defects and disordered carbon atoms, and the G band at 1590 cm^{-1} corresponds to the stretching vibration of sp^2 hybridized carbon atoms in graphite [36]. For $\text{MnFe}_2\text{O}_4@\text{SBC}$, the I_D/I_G ratio was 0.99, which is slightly higher than MnFe_2O_4 ($I_D/I_G = 0.98$) and significantly lower than SBC ($I_D/I_G = 1.03$). The decrease in the I_D/I_G ratio for $\text{MnFe}_2\text{O}_4@\text{SBC}$ indicates a reduction in disorder and structural defects compared to SBC, which improved the material catalytic performance [35]. In addition, high degree of graphitization in $\text{MnFe}_2\text{O}_4@\text{SBC}$ facilitated the exposure of active sites on the catalyst surface [37]. The structural defects and degree of graphitization of $\text{MnFe}_2\text{O}_4@\text{SBC}$ prepared at different calcination temperatures were also evaluated. As shown in Figure 2d, the I_D/I_G value decreased from 1.24 for $\text{MnFe}_2\text{O}_4@\text{SBC}$ -350 to 0.98 for $\text{MnFe}_2\text{O}_4@\text{SBC}$ -550, indicating that increasing the calcination temperature enhanced graphitization of the carbon framework. Interestingly, the I_D/I_G value slightly increased to 0.99 for $\text{MnFe}_2\text{O}_4@\text{SBC}$ -750, suggesting the generation of additional edge defects and structural regulation during the carbon reconstruction process [26]. When the calcination temperature further increased to 950°C , the I_D/I_G value decreased to 0.95, indicating that excessive thermal treatment promoted the formation of ordered sp^2 carbon structures. Notably, $\text{MnFe}_2\text{O}_4@\text{SBC}$ -750 maintained a relatively high defect density while simultaneously developing an extended π -conjugated structure. This structural feature is favorable for enhancing interfacial π -d electronic coupling with MnFe_2O_4 , thereby facilitating electron transfer and promoting PI activation (discussed in Section 3.4).

The surface chemical states of $\text{MnFe}_2\text{O}_4@\text{SBC}$ were examined through XPS. As shown in Figure 2e, the XPS survey spectrum showed clear signals of C, O, Mn and Fe, which was consistent with the expected composition of $\text{MnFe}_2\text{O}_4@\text{SBC}$. The C 1s spectrum displayed four deconvoluted peaks at 284.80 eV (C-C), 286.58 eV (C-O), 288.23 eV (C=O) and 290.51 eV (O-C=O), suggesting that oxygen-containing groups remained abundant on the composite surface (Figure 2f) [38,39]. The O 1s spectrum showed characteristic peaks at 530.00 eV, corresponding to lattice oxygen species such as Mn-O and Fe-O, while peaks at 531.63 eV, 533.34 eV and 535.04 eV were assigned to C-O, O-C=O and C=O, respectively (Figure 2g) [39,40]. The results were consistent with the findings from the XRD and FTIR analyses. The coexistence of lattice oxygen and surface oxygenated groups implied that the biochar support influenced the coordination environment of Mn and Fe, thereby enhancing the surface reactivity. The Mn 2p spectrum contained three Mn 2p_{3/2} peaks at 641.66 eV (Mn^{2+}), 643.94 eV (Mn^{3+}) and 646.23 eV (Mn^{4+}) (Figure 2h). Several peaks between 652 and 658 eV corresponded to Mn 2p_{1/2} [41,42]. The simultaneous presence of Mn^{2+} , Mn^{3+} and Mn^{4+} suggested a mixed-valence Mn environment, which was attributed to facilitate rapid electron exchange. The Fe 2p spectrum further supported the mixed-valence nature of the composite. Peaks at 711.79 eV (Fe^{2+} 2p_{3/2}) and 714.50 eV (Fe^{3+} 2p_{3/2}) were observed, along with a satellite peak at 719.62 eV (Figure 2i) [28]. The Fe 2p_{1/2} region presented peaks at 724.74 eV (Fe^{2+}), 727.78 eV (Fe^{3+}), and 731.16 eV for the satellite feature. The

coexistence of Fe^{2+} and Fe^{3+} indicated that Fe sites could participate in redox cycling, which is essential for sustaining catalytic activity. Additionally, the XPS analysis of SBC was presented in Text S6. Compared with $\text{MnFe}_2\text{O}_4@\text{SBC}$, pristine SBC exhibited weaker Fe signals, and the Mn 2p spectrum could not be reliably deconvoluted due to the low Mn content. The metal species in SBC were mainly present as dispersed and less-defined oxides. In contrast, $\text{MnFe}_2\text{O}_4@\text{SBC}$ displayed significantly enhanced Mn and Fe signals with well-resolved mixed-valence states, indicating the formation of well-defined MnFe_2O_4 phases and a reconstructed coordination environment.

3.2. Catalytic Performance Towards BPA Removal

The effect of different MnFe_2O_4 : SBC ratios on BPA degradation was studied to evaluate the catalytic performance of the $\text{MnFe}_2\text{O}_4@\text{SBC}$ system (Figures 3a and S3a). At a 1:1 MnFe_2O_4 :SBC ratio, the system achieved the highest catalytic efficiency, with a BPA removal rate of 98.8% ($k_{\text{obs}} = 0.147 \text{ min}^{-1}$). When the ratio increased to 1:3, the BPA removal rate slightly decreased to 98.6%, with a k_{obs} value of 0.136 min^{-1} . Further increasing the SBC content to 1:5 resulted in a slight decrease in BPA removal efficiency to 97.8%, with a corresponding k_{obs} of 0.133 min^{-1} . The results indicated that although the system experienced a slight decline in performance, it still maintained high catalytic potential. At a 1:7 ratio, the BPA removal rate dropped to 88.3%, with a k_{obs} of 0.064 min^{-1} . At a 1:9 and 1:11 ratio, the removal efficiency further decreased to 83.1% and 69.1%, and the k_{obs} was reduced to 0.052 min^{-1} and 0.033 min^{-1} . These findings suggested that while increasing the SBC content improved the structural stability of the catalyst, an excessively low MnFe_2O_4 :SBC ratio resulted in insufficient surface-active sites. Therefore, a MnFe_2O_4 :SBC ratio of 1:3 was selected as the optimal catalyst, as it provided sufficient active sites for efficient PI activation and maximized BPA degradation efficiency.

The BPA degradation efficiency was evaluated using $\text{MnFe}_2\text{O}_4@\text{SBC}$ catalysts prepared from SBC calcined at different temperatures (Figures 3b and S3b). $\text{MnFe}_2\text{O}_4@\text{SBC}$ -350, $\text{MnFe}_2\text{O}_4@\text{SBC}$ -550, and $\text{MnFe}_2\text{O}_4@\text{SBC}$ -750 removed 58.3%, 77.7% and 98.6% of BPA in solution within 30 min, respectively, with corresponding k_{obs} values of 0.024 min^{-1} , 0.048 min^{-1} and 0.136 min^{-1} . These results indicated that increasing the calcination temperature promoted the formation of more ordered sp^2 carbon structures, which facilitated interfacial electron transfer between the carbon framework and MnFe_2O_4 [43]. Notably, $\text{MnFe}_2\text{O}_4@\text{SBC}$ -750 exhibited the highest degradation efficiency, which was attributed to the coexistence of abundant defect sites and a relatively developed graphitic structure. However, the BPA degradation efficiency in the $\text{MnFe}_2\text{O}_4@\text{SBC}$ -950/PI system decreased to 85.8% ($k_{\text{obs}} = 0.064 \text{ min}^{-1}$), indicating that excessively high pyrolysis temperatures led to over-graphitization, thereby weakening the surface reactivity. Therefore, $\text{MnFe}_2\text{O}_4@\text{SBC}$ -750 was selected for further investigation.

The degradation performance of BPA by $\text{MnFe}_2\text{O}_4@\text{SBC}/\text{PI}$, $\text{MnFe}_2\text{O}_4/\text{PI}$, SBC/PI and PI alone was compared to assess the catalytic efficiency of the different systems. To avoid exaggerating the degradation efficiency of the catalytic process, a 30 min pre-adsorption (without PI addition) was performed. The results showed that BPA removal through adsorption in all systems did not exceed 15%, indicating that the contribution of the adsorption process to pollutant removal was limited. As illustrated in Figures 3c and S3c, The $\text{MnFe}_2\text{O}_4@\text{SBC}/\text{PI}$ system demonstrated the highest degradation efficiency, with a BPA removal rate of 98.6% and a corresponding pseudo-first-order kinetics (k_{obs}) constant of 0.136 min^{-1} . In contrast, the $\text{MnFe}_2\text{O}_4/\text{PI}$ and SBC/PI systems showed lower degradation rates, with values of approximately 28.8% ($k_{\text{obs}} = 0.010 \text{ min}^{-1}$) and 21.7% ($k_{\text{obs}} = 0.006 \text{ min}^{-1}$), respectively. Notably, the PI-only system exhibited negligible removal, confirming that PI alone has limited degradation capability for BPA. The results

indicated that $\text{MnFe}_2\text{O}_4@\text{SBC}$ exhibits significantly superior catalytic performance for PI activation compared to other systems. Moreover, Text S7 analyzed the effects of catalyst dosage, PI concentration, BPA concentration and initial pH on the BPA removal efficiency of the $\text{MnFe}_2\text{O}_4@\text{SBC}/\text{PI}$ system. Considering catalytic activity and PI activation efficiency, the optimal conditions selected were $[\text{MnFe}_2\text{O}_4@\text{SBC}] = 0.4 \text{ g L}^{-1}$, $[\text{BPA}] = 20 \text{ mg L}^{-1}$, $[\text{PI}] = 0.5 \text{ mM}$, and an initial pH of 7.

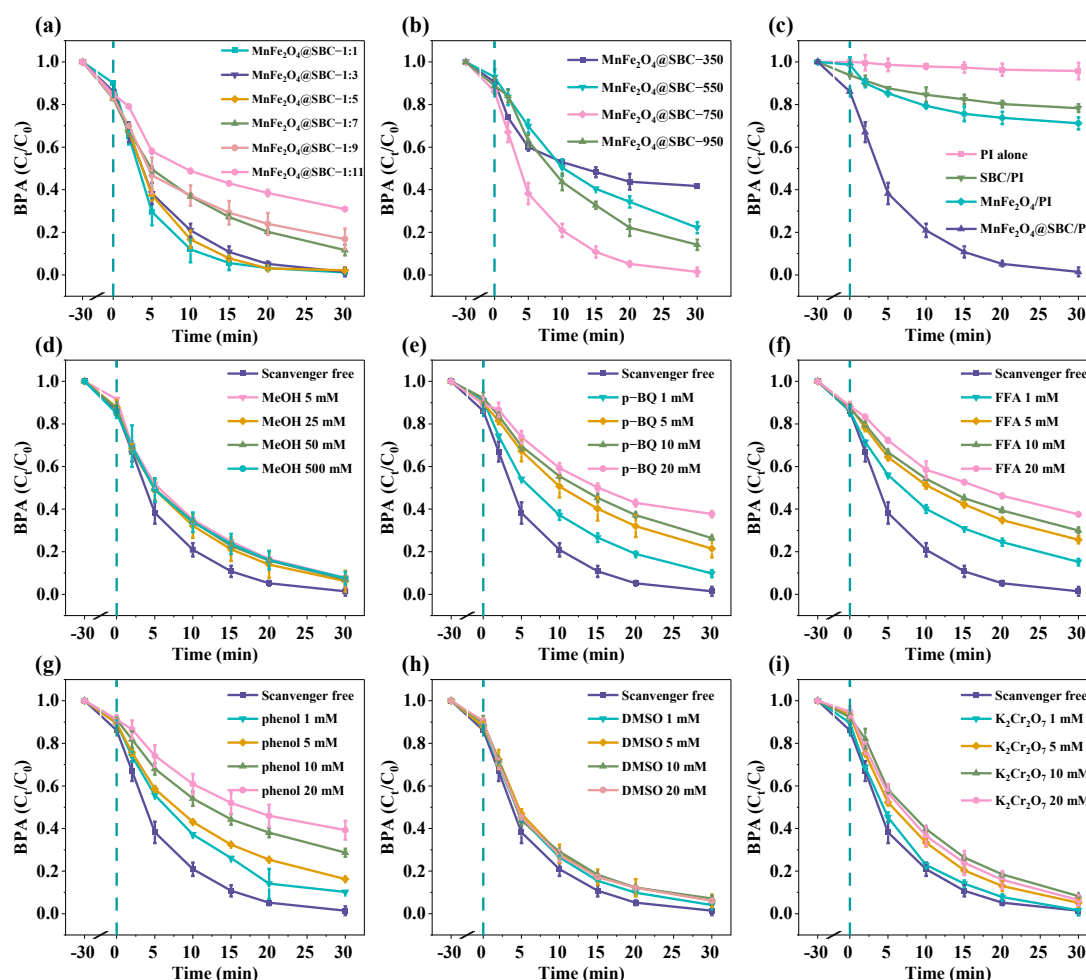


Figure 3. (a) Effect of MnFe_2O_4 loading ratio on the BPA degradation performance of $\text{MnFe}_2\text{O}_4@\text{SBC}$, (b) effect of calcination temperature on the catalytic performance of $\text{MnFe}_2\text{O}_4@\text{SBC}$, (c) comparison of BPA removal efficiency in different systems, effect of different quenchers on BPA degradation in the $\text{MnFe}_2\text{O}_4@\text{SBC}/\text{PI}$ system, including (d) MeOH, (e) p-BQ, (f) FFA, (g) phenol, (h) DMSO and (i) $\text{K}_2\text{Cr}_2\text{O}_7$.

3.3. Identification of ROS

To clarify the reactive species responsible for BPA degradation in the $\text{MnFe}_2\text{O}_4@\text{SBC}/\text{PI}$ system, a series of quenching experiments was conducted using different scavengers at varied concentrations. Methanol (MeOH) ($k_{\text{obs}} = (3.8\text{--}7.6) \times 10^8 \text{ M}^{-1} \text{ s}^{-1}$) can be an $\bullet\text{OH}$ quencher [44]. As shown in Figures 3d and S4a, the addition of 5–500 mM MeOH caused only slight decrease in BPA removal (92.3–96.7%), with k_{obs} ranging from 0.087 to 0.079 min^{-1} , indicating that $\bullet\text{OH}$ contributed minimally to the oxidation process. p-Benzoquinone (p-BQ) was introduced to examine the contribution of $\text{O}_2^{\bullet-}$ in the reaction system ($k_{\text{obs}} = (0.9\text{--}1.0) \times 10^9 \text{ M}^{-1} \text{ s}^{-1}$) [45]. With p-BQ concentrations of 1, 5, 10 and 20 mM, BPA removal decreased to 90.2%, 78.5%, 73.6% and 62.3%, with k_{obs} of 0.073, 0.048, 0.042 and 0.031 min^{-1} , suggesting that $\text{O}_2^{\bullet-}$ played a significant role in BPA degradation

(Figures 3e and S4b). Furfuryl alcohol (FFA) preferentially scavenges $^1\text{O}_2$ in the system ($k_{\text{obs}} = 1.2 \times 10^8 \text{ M}^{-1} \text{ s}^{-1}$) [46]. The results showed that FFA also suppressed BPA degradation, with removal efficiencies dropping from 98.6% to 62.5% and k_{obs} decreasing from 0.136 to 0.029 min^{-1} as the FFA concentration increased, confirming the participation of $^1\text{O}_2$ (Figures 3f and S4c). Notably, the quenching effect of FFA on $^1\text{O}_2$ may be overestimated, because FFA can directly consume PI through an electron-transfer reaction. Therefore, FFA was used in place of BPA to examine the possible competitive consumption of PI in the system. As shown in Figure S5, the PI concentration remained unchanged in the presence of 20 mM FFA, indicating that PI consumption by FFA was negligible. Phenol was selected as the quenching agent for $\text{IO}_3^\bullet$, and its addition markedly inhibited the removal of BPA ($k_{\text{obs}} = 1.0 \times 10^9 \text{ M}^{-1} \text{ s}^{-1}$) [47]. BPA removal rate decreased from 98.6% ($k_{\text{obs}} = 0.136 \text{ min}^{-1}$) to 89.8% ($k_{\text{obs}} = 0.076 \text{ min}^{-1}$), 83.9% ($k_{\text{obs}} = 0.067 \text{ min}^{-1}$), 71.3% ($k_{\text{obs}} = 0.039 \text{ min}^{-1}$) and 60.7% ($k_{\text{obs}} = 0.029 \text{ min}^{-1}$) when phenol concentrations increased from 0 mM to 1 mM, 5 mM, 10mM and 20 mM, respectively, indicating that $\text{IO}_3^\bullet$ was one of the dominant reactive species in this system (Figures 3g and S4d). 2,4,6-TCP was employed as a target contaminant to further verify the contribution of $\text{IO}_3^\bullet$ in the $\text{MnFe}_2\text{O}_4/\text{SBC}/\text{PI}$ system, because $\text{IO}_3^\bullet$ is unable to oxidize 2,4,6-TCP. The degradation efficiency of 2,4,6-TCP reached 52.7% within 14 min, confirming the presence of $\text{IO}_3^\bullet$ in the $\text{MnFe}_2\text{O}_4/\text{SBC}/\text{PI}$ system (Figure S6). Since SBC contains multiple metals and supports MnFe_2O_4 , dimethyl sulfoxide (DMSO) was employed to exclude the possible involvement of high-valent metal-oxo species. BPA degradation remained above 90%, demonstrating that high-valent metal-oxo species were not involved (Figures 3h and S4e). $\text{K}_2\text{Cr}_2\text{O}_7$ was employed as an electron scavenger to determine whether ROS generation in the system involves solution phase electron transfer pathways [48]. As shown in Figures 3i and S4f, $\text{K}_2\text{Cr}_2\text{O}_7$ caused negligible inhibition of BPA concentration, with BPA removal efficiencies of 91.7–98.4% and k_{obs} values of $0.081\text{--}0.131 \text{ min}^{-1}$, suggesting that the active sites on the catalyst surface directly activate the adsorbed PI to generate ROS. The premixing experiment was used to identify ROS generation. When $\text{MnFe}_2\text{O}_4/\text{SBC}$ and PI were premixed for 5, 10 and 20 min before BPA addition, the BPA removal efficiencies decreased to 62.3%, 42.4% and 19.2%, respectively (Figure S7). These results indicated that with increasing premixing time, the ROS produced in the system underwent self-quenching and were no longer available to effectively degrade BPA. Electrochemical measurements were employed to examine whether a direct electron-transfer pathway existed in the $\text{MnFe}_2\text{O}_4/\text{SBC}/\text{PI}$ system. The results demonstrated that BPA did not transfer electrons to IO_4^- via the catalyst surface, indicating that a direct electron-transfer process was absent (Text S8).

EPR spectroscopy, using 5,5-dimethyl-1-pyrroline-N-oxide (DMPO) and 2,2,6,6-tetramethylpiperidine (TEMP) as spin-trapping agents, was performed to identify the ROS generated in the $\text{MnFe}_2\text{O}_4/\text{SBC}/\text{PI}$ system (Figure 4). No EPR signals were detected in either the $\text{MnFe}_2\text{O}_4/\text{SBC}$ or PI-only systems, confirming that neither component produced measurable radicals in isolation. As expected, when $\text{MnFe}_2\text{O}_4/\text{SBC}$ was coupled with PI, distinct EPR signals were observed. At 5 min of reaction, $\text{DMPO}\cdot\text{OH}$, $\text{DMPO}\cdot\text{O}_2^{\bullet-}$ and $\text{DMPO}\cdot\text{IO}_3^\bullet$ were detected. These signals became more pronounced at 10 min, demonstrating sustained radical generation within the system (Figure 4a–c). Specifically, the $\text{DMPO}\cdot\text{OH}$ signal showed a characteristic quartet pattern (1:2:2:1), while the $\text{DMPO}\cdot\text{O}_2^{\bullet-}$ signal appeared as a quartet with a 1:1:1:1 intensity ratio [45,49]. The $\text{DMPO}\cdot\text{IO}_3^\bullet$ adduct exhibited a signal consistent with a quartet (2:2:1:2:1:2), confirming the generation of $\text{IO}_3^\bullet$ in the reaction [47]. Furthermore, TEMP was used to capture $^1\text{O}_2$. A distinct three-line EPR signal with a 1:1:1 intensity ratio ($\text{TEMP}\cdot^1\text{O}_2$) appeared at 5 min and the signal intensity increased at 10 min, indicating that $^1\text{O}_2$ was generated during the reaction (Figure 4d) [50]. Upon adding BPA to the system, a noticeable decrease in signal intensity was observed

for all four signals, indicating that the primary reactive species in the $\text{MnFe}_2\text{O}_4@\text{SBC}/\text{PI}$ system were $\bullet\text{OH}$, $\text{O}_2^{\bullet-}$, $\text{IO}_3^\bullet$ and $^1\text{O}_2$.

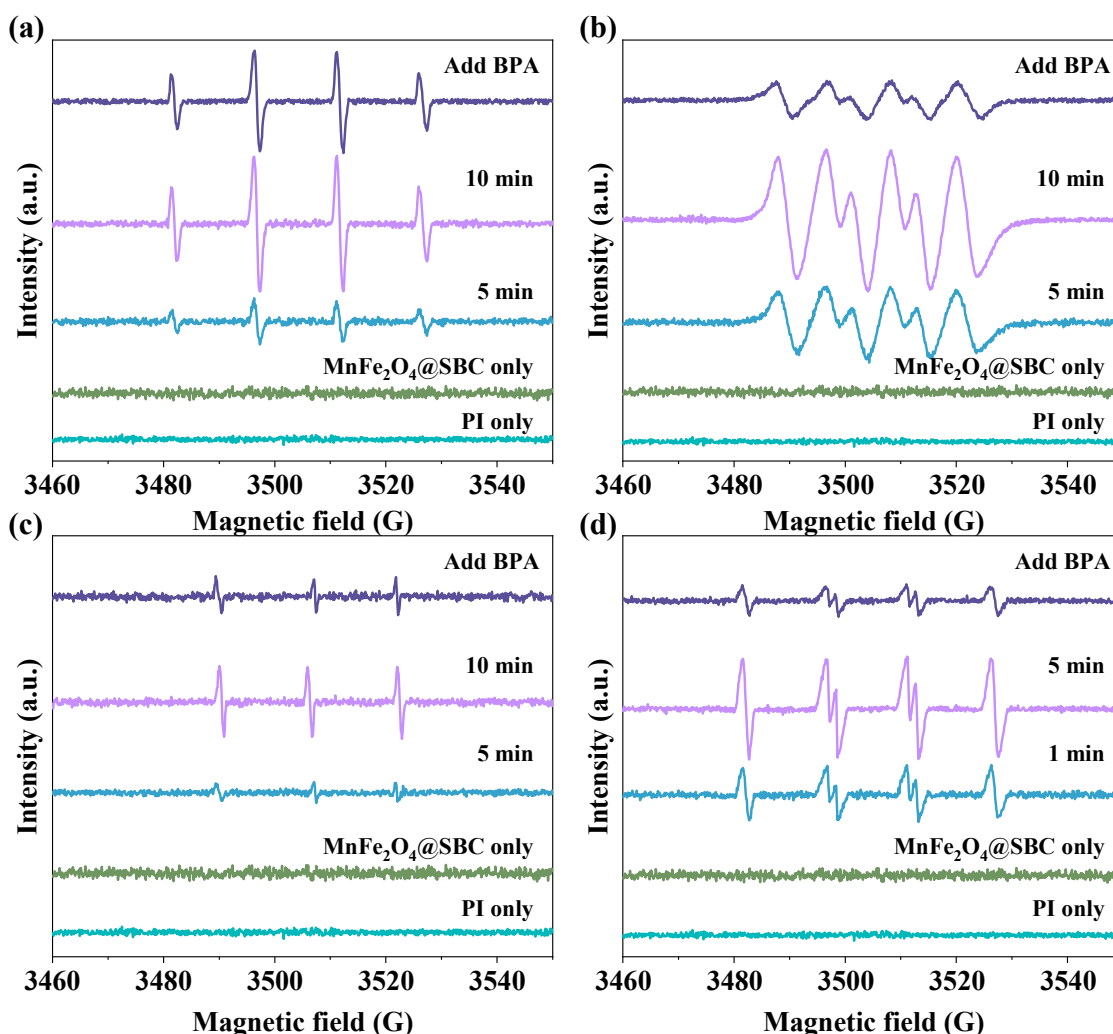
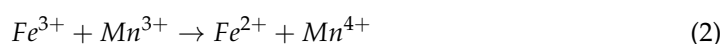
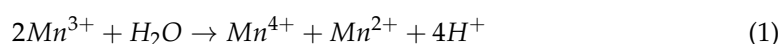


Figure 4. EPR spectra of (a) $\text{DMPO}\cdot\text{OH}$, (b) $\text{DMPO}\cdot\text{O}_2^{\bullet-}$, (c) $\text{TEMP}\cdot^1\text{O}_2$ and (d) $\text{DMPO}\cdot\text{IO}_3^\bullet$ in PI alone, $\text{MnFe}_2\text{O}_4@\text{SBC}$ alone, $\text{MnFe}_2\text{O}_4@\text{SBC}/\text{PI}$ and $\text{MnFe}_2\text{O}_4@\text{SBC}/\text{PI}$ with BPA.

3.4. Catalytic Mechanism of $\text{MnFe}_2\text{O}_4@\text{SBC}$

The XPS analysis was employed to investigate the changes in chemical bonds and functional groups on the $\text{MnFe}_2\text{O}_4@\text{SBC}$ surface before and after the reaction (Figure S8). The C1s spectrum showed that before the reaction, the relative contents of C-C, C-O, and C=O were 46.0%, 40.4% and 13.6%, respectively. After the reaction, the relative content of these peaks shifted slightly to 46.6%, 41.0% and 12.4%, respectively. These results suggested that while the surface carbon structure remained largely unchanged, there was a shift in the carbon bonding environment, possibly due to the adsorption of reaction intermediates or surface oxidation during the catalytic process. The O1s spectrum (Figure S8a) showed that before the reaction, the relative contents of surface oxygen species C-O, O-C=O, and C=O were 35.2%, 40.3% and 8.1%, respectively. After the reaction, the relative content of C-O and O-C=O increased slightly to 38.6% and 42.1%, respectively, while the content of C=O remained. Notably, the M-O peak decreased from 16.4% to 10.7%, suggesting that M-O was a potential active site for PI activation. The Mn 2p spectra revealed that before the reaction, the Mn species were distributed as Mn^{2+} (37.7%), Mn^{3+} (40.8%) and Mn^{4+} (21.5%) (Figure S8b). After the reaction, the Mn^{2+} content increased to 39.6%, while the Mn^{3+} content decreased to 36.0%. In contrast, the Mn^{4+} content increased to 28.4%.

Similarly, the relative content of Fe^{2+} and Fe^{3+} was 50.2% and 29.7% before the reaction, respectively (Figure S8c,d). After the reaction, the content of Fe^{2+} decreased to 36.6%, while Fe^{3+} increased to 45.5%. These changes suggested that during the catalytic degradation process, Mn^{2+} (Fe^{2+}) and Mn^{3+} (Fe^{3+}) were involved in the redox cycling, leading to the interconversion of Mn species. The oxidation of Mn^{2+} to Mn^{3+} and Mn^{4+} , as well as the oxidation of Fe^{2+} to Fe^{3+} , facilitated the generation of ROS. However, the slight increase in Mn^{2+} content after the reaction was due to the regeneration of Mn^{2+} from Mn^{3+} through reaction with water (Equation (1)) [51]. Furthermore, since the standard reduction potential of $\text{Fe}^{3+}/\text{Fe}^{2+}$ (0.77 V) was higher than that of $\text{Mn}^{4+}/\text{Mn}^{3+}$ (0.15 V), Fe^{3+} tended to be reduced to Fe^{2+} by Mn^{3+} (Equation (2)) [14]. The redox cycling of $\text{Fe}^{3+}/\text{Fe}^{2+}$ and $\text{Mn}^{4+}/\text{Mn}^{3+}$ regenerated Fe^{2+} , which also explained the increased content of Mn^{4+} after the reaction.



DFT calculations were conducted to elucidate the influence of SBC calcination temperature on the interfacial interaction and electron transfer behavior of $\text{MnFe}_2\text{O}_4@\text{SBC}$ toward PI. As the calcination temperature increased from 350 to 750 °C, the I-O bonds in PI adsorbed on $\text{MnFe}_2\text{O}_4@\text{SBC}$ were progressively elongated from 2.68 Å ($\text{MnFe}_2\text{O}_4@\text{SBC}$ -350) to 3.87 Å ($\text{MnFe}_2\text{O}_4@\text{SBC}$ -550) and further to 4.01 Å ($\text{MnFe}_2\text{O}_4@\text{SBC}$ -750), indicating a continuous enhancement in PI activation (Figure 5a–c). Correspondingly, the calculated adsorption energies (E_{ads}) became more negative, increasing from −7.22 eV to −10.85 eV and reaching a maximum of −12.76 eV on $\text{MnFe}_2\text{O}_4@\text{SBC}$ -750, suggesting the formation of a highly stable adsorption configuration. In contrast, further increasing the calcination temperature to 950 °C led to a decrease in both I-O bond elongation (3.82 Å) and E_{ads} (−8.71 eV), implying weakened interfacial interaction (Figure 5d). As shown in Figure 5e–h, the charge transfer values of $\text{MnFe}_2\text{O}_4@\text{SBC}$ -350 (0.98 e) and $\text{MnFe}_2\text{O}_4@\text{SBC}$ -550 (1.55 e) were significantly lower than that of $\text{MnFe}_2\text{O}_4@\text{SBC}$ -750 (1.68 e), which can be attributed to excessive structural disorder at lower temperatures that limits π -electron delocalization and suppresses electron transport [52]. Meanwhile, $\text{MnFe}_2\text{O}_4@\text{SBC}$ -950 exhibited a reduced Bader charge transfer (1.02 e) compared with $\text{MnFe}_2\text{O}_4@\text{SBC}$ -750, indicating that excessive graphitization stabilizes π electrons and weakens interfacial electronic coupling with MnFe_2O_4 . Notably, $\text{MnFe}_2\text{O}_4@\text{SBC}$ -750 achieves an optimal balance between framework defects and graphitic domains. In this structure, defect-induced electron enrichment at interfacial sites enhances electronic polarization and reactivity, thereby facilitating efficient electron transfer from the catalyst to PI. This synergistic regulation of defect density and electronic structure explains the superior PI activation capability of $\text{MnFe}_2\text{O}_4@\text{SBC}$ -750.

The electronic structures of $\text{MnFe}_2\text{O}_4@\text{SBC}$ prepared at different calcination temperatures were further analyzed by total and projected density of states (TDOS and PDOS) calculations to elucidate the role of SBC structure in regulating interfacial electron transfer toward PI. As shown in Figure 5i,j, $\text{MnFe}_2\text{O}_4@\text{SBC}$ -350 and $\text{MnFe}_2\text{O}_4@\text{SBC}$ -550 exhibited a relatively low TDOS near the Fermi level, indicating limited availability of charge carriers on the surface of catalyst [53]. Although abundant structural defects were generated at low calcination temperature, excessive disorder broke the continuity of the π -conjugated framework. This led to strong localization of C 2p electrons and inhibited efficient electron transport (Figure 5m,n). Notably, $\text{MnFe}_2\text{O}_4@\text{SBC}$ -750 exhibited the highest TDOS around the Fermi level, together with pronounced hybridization between C 2p and Mn/Fe 3d states (Figure 5k,o). This strong orbital overlap demonstrated the formation of π -d electronic coupling interface, which facilitated rapid electron delocalization across the carbon support and electron transfer from catalyst to the adsorbed PI [54]. In contrast, $\text{MnFe}_2\text{O}_4@\text{SBC}$ -950

exhibited stabilized π electrons within the carbon framework due to excessive graphitization, weakening π -d electronic coupling with MnFe_2O_4 and suppressing electron transfer to the adsorbed PI (Figure 5l,p).

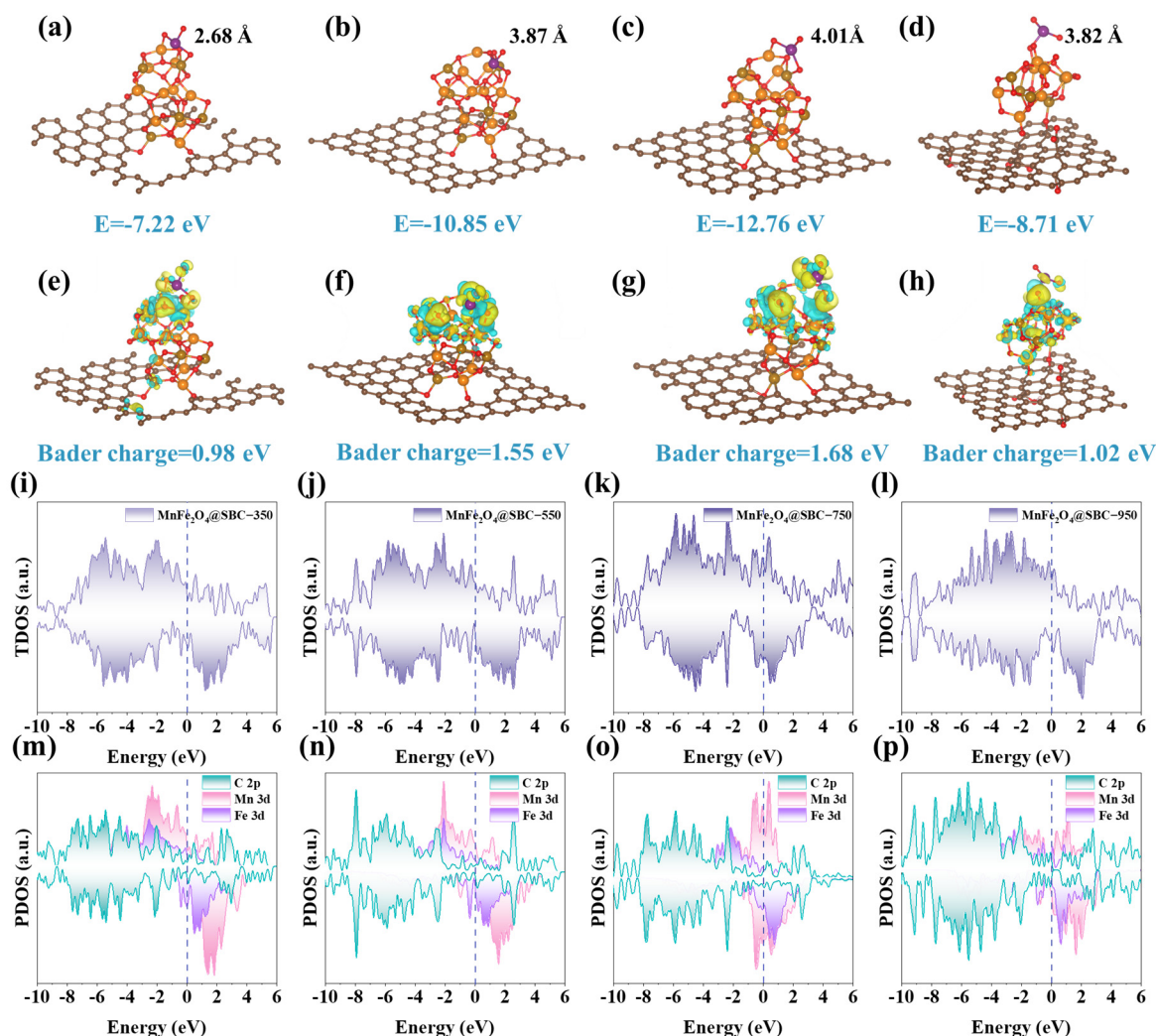


Figure 5. (a–d) Optimized adsorption geometries of IO_4^- on $\text{MnFe}_2\text{O}_4@\text{SBC}-x$ with the corresponding E_{ads} , (e–h) charge density difference distributions of IO_4^- adsorbed on $\text{MnFe}_2\text{O}_4@\text{SBC}_x$, where yellow and cyan denote electron accumulation and depletion, (i–l) TDOS and (m–p) PDOS of $\text{MnFe}_2\text{O}_4@\text{SBC}_x$ ($E_f = 0$ eV).

DFT calculations were further performed to examine the interaction mechanism of PI with SBC, MnFe_2O_4 and $\text{MnFe}_2\text{O}_4@\text{SBC}$. In the SBC/PI system, the four I-O bonds of PI remained relatively short (1.90 Å, 1.76 Å, 1.78 Å and 1.78 Å), indicating only weak perturbation of the oxidant structure. When PI was adsorbed on MnFe_2O_4 , the I-O bonds were 2.77 Å, 2.34 Å, 2.28 Å and 2.16 Å, suggesting that MnFe_2O_4 possessed partial activation capability toward PI. Notably, in the $\text{MnFe}_2\text{O}_4@\text{SBC}/\text{PI}$ system, the I-O bonds were markedly elongated to 3.87 Å, 1.92 Å, 1.91 Å and 1.90 Å, demonstrating that the I-O bonds become more susceptible to cleavage and subsequent activation of PI. The E_{ads} of PI on SBC, MnFe_2O_4 and $\text{MnFe}_2\text{O}_4@\text{SBC}$ were calculated to be -0.79 eV, -4.53 eV and -10.85 eV, respectively. The significantly more favorable adsorption on $\text{MnFe}_2\text{O}_4@\text{SBC}$ indicated the formation of a highly stable adsorption configuration, arising from enhanced electronic coupling between MnFe_2O_4 and SBC. Bader charge analysis further revealed that electron transfer from the catalysts to PI followed the order SBC (0.45 e) < MnFe_2O_4 (0.92 e) < $\text{MnFe}_2\text{O}_4@\text{SBC}$ (1.58 e). The strong electron-donating properties of $\text{MnFe}_2\text{O}_4@\text{SBC}$

enhanced interfacial electronic polarization, thereby promoting the cleavage of the I-O bonds.

Overall, as illustrated in Figure 6, the PI activation performance of $\text{MnFe}_2\text{O}_4@\text{SBC}$ originates from the synergistic regulation of defect structure, graphitization degree and interfacial electronic coupling. The optimized SBC framework constructed at appropriate calcination temperature provides abundant defect sites and a continuous π -conjugated network, which promotes π -electron delocalization and facilitates rapid interfacial electron transfer. Meanwhile, the strong π -d orbital coupling between C 2p and Mn/Fe 3d states enhances electronic polarization and accelerates electron migration from $\text{MnFe}_2\text{O}_4@\text{SBC}$ to adsorbed PI, thereby weakening and elongating the I-O bonds of PI. In addition, the redox cycling of $\text{Mn}^{2+}/\text{Mn}^{3+}/\text{Mn}^{4+}$ and $\text{Fe}^{2+}/\text{Fe}^{3+}$ further promotes ROS generation and sustains catalytic activity.

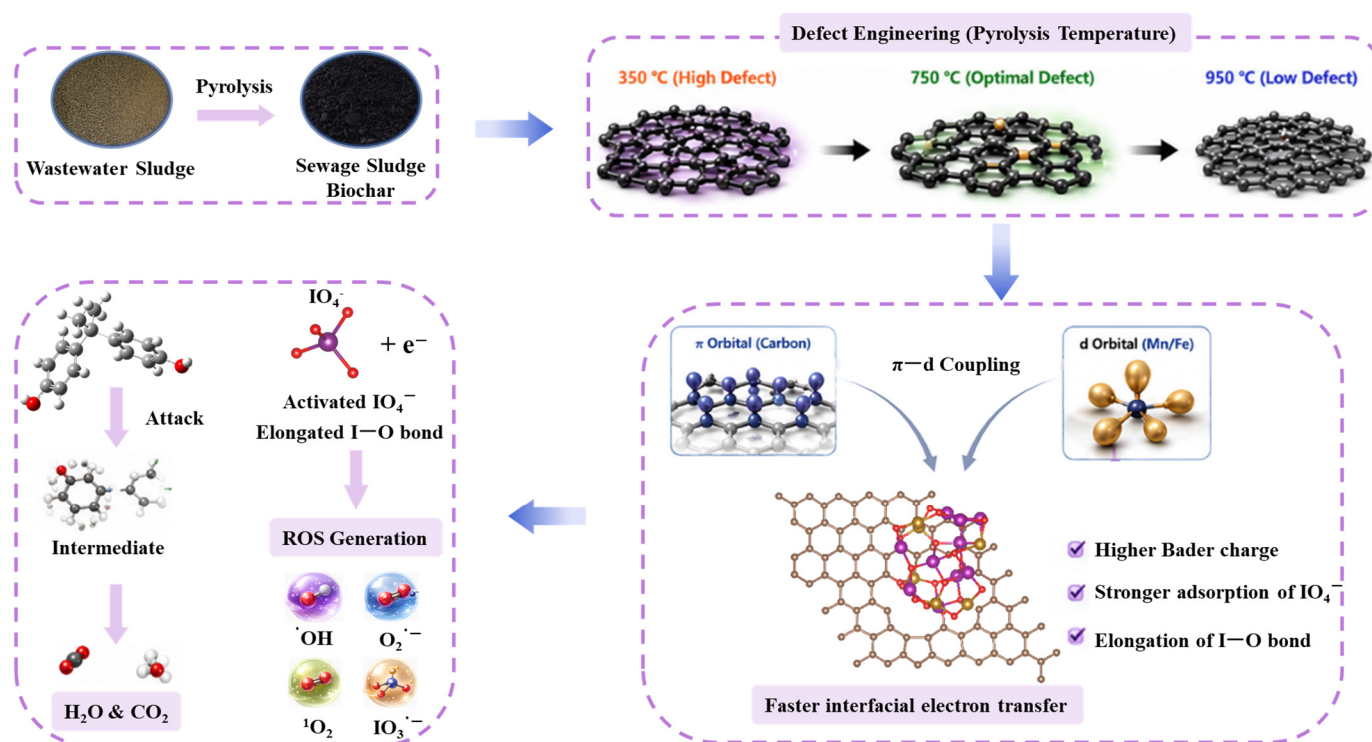


Figure 6. Mechanism of BPA degradation in $\text{MnFe}_2\text{O}_4@\text{SBC}/\text{PI}$.

3.5. Degradation Paths of BPA in the $\text{MnFe}_2\text{O}_4@\text{SBC}/\text{PI}$ System

To further clarify the reactive regions of BPA in the $\text{MnFe}_2\text{O}_4@\text{SBC}/\text{PI}$ system, frontier molecular orbital analysis was performed using DFT calculations. As shown in Figure 7a, the highest occupied molecular orbitals (HOMO) were predominantly localized on the aromatic rings and the phenolic oxygen atoms, indicating that these electron-rich domains served as the primary electron-donating sites during oxidative reactions [55]. In contrast, the lowest unoccupied molecular orbitals (LUMO) were mainly distributed over several carbon atoms on the benzene rings, suggesting that these positions function as favorable electron-accepting sites [56]. The calculated HOMO (-6.024 eV) and LUMO (-0.213 eV) levels resulted in an energy gap of 5.811 eV, reflecting a relatively stable molecular structure. However, such a gap still permits electrophilic, nucleophilic and radical attacks when strong oxidizing species ($\cdot\text{OH}$, $\text{O}_2^{\cdot-}$, $\text{IO}_3^{\cdot}$ and $^1\text{O}_2$) were present in the $\text{MnFe}_2\text{O}_4@\text{SBC}/\text{PI}$ system. The electrostatic potential (ESP) surface further showed strong negative potential around the phenolic oxygens and nearby ortho-carbons, indicating that these electron-rich sites were susceptible to electrophilic oxidation (Figure 7b). The relatively neutral to slightly

positive regions located on the isopropylidene bridge and para-positions of the aromatic rings provide additional pathways for nucleophilic or radical attacks [57].

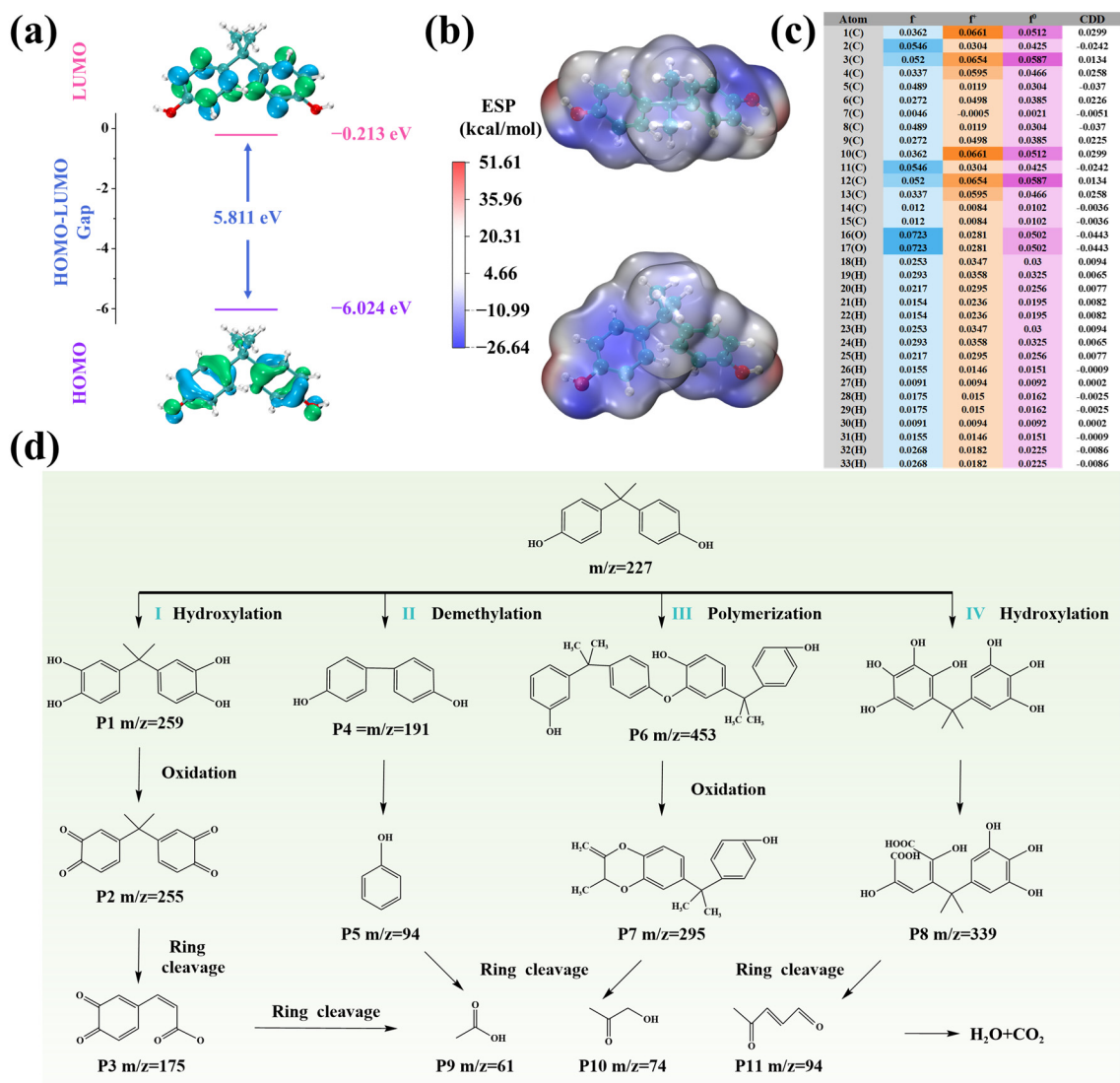


Figure 7. (a) LUMO and HOMO distribution of BPA and HOMO-LUMO gap, (b) top view and front view of the ESP mapping of BPA, (c) Fukui index f^- , f^+ and f^0 of BPA and (d) degradation pathways.

In addition, the condensed Fukui functions were calculated to identify atom-specific reactivity toward electrophilic (f^-), radical (f^0), and nucleophilic (f^+) species (Figures 7c and S9). Atoms O16, O17, C2, C3, C11 and C12 exhibited relatively high f^- values, suggesting that these positions were particularly vulnerable to electrophilic oxidants [58]. Meanwhile, atoms C1, C3, C10, C12, O16 and O17 showed elevated f^0 values, indicating their pronounced reactivity toward radical species such as $\bullet OH$ and $IO_3\bullet$ [59]. Furthermore, the f^+ values were higher at C1, C10, C3, C12, C4 and C13, indicating that these positions are more susceptible to nucleophilic radical attack by $O_2^{\bullet-}$ [60]. The condensed dual descriptor (CDD) analysis revealed large absolute values around the phenolic hydroxyl groups and aromatic carbons (O16, O17, C5, C8, C1 and C10), confirming that these regions are the dominant reactive centers during BPA transformation [61]. These results demonstrated that the oxidative degradation of BPA over $MnFe_2O_4@SBC$ was strongly governed by electrophilic and radical attack at electron-rich oxygen and carbon sites, consistent with the experimentally observed degradation behavior.

To elucidate intermediate products of BPA in the $\text{MnFe}_2\text{O}_4@\text{SBC}/\text{PI}$ system, the intermediates detected by LC-MS were used to propose four possible degradation pathways, as shown in Figures 7d and S10. In Pathway I, $\bullet\text{OH}$ and $\text{IO}_3\bullet$ preferentially targeted the electron-rich carbon sites adjacent to the phenolic hydroxyl groups of BPA molecule (C1, C3, C10, C12), producing the hydroxylated intermediate P1 ($m/z = 259$). Subsequent multi-step oxidation of P1 yielded the quinone-type species P2 ($m/z = 255$), which then underwent oxidative ring-opening to generate P3 ($m/z = 175$) [62]. Pathway II involved $^1\text{O}_2$ mediated oxidative demethylation of BPA, forming P4 ($m/z = 191$). Further oxidation of P4 led to cleavage of the isopropylidene bridge, producing low-molecular-weight phenolic fragments P5 ($m/z = 94$) [63]. In Pathway III, BPA underwent radical-induced polymerization followed by oxidative coupling, generating a dimeric intermediate P6 ($m/z = 453$). Progressive oxidation of this dimer produced P7 ($m/z = 295$), indicating that polymerization–depolymerization cycles contributed to BPA degradation [64]. Pathway IV proceeded through direct hydroxylation and aromatic ring scission to generate P8 ($m/z = 339$) [65]. Ultimately, P3, P5, P7 and P8 were further attacked by ROS to undergo ring-opening and fragmentation reactions, producing small organic acids and low-molecular-weight compounds P9 ($m/z = 61$), P10 ($m/z = 74$) and P11 ($m/z = 94$) [66,67]. These species were subsequently mineralized to CO_2 and H_2O .

3.6. Stability and Practical Applicability of $\text{MnFe}_2\text{O}_4@\text{SBC}$

The durability and practical applicability of the $\text{MnFe}_2\text{O}_4@\text{SBC}/\text{PI}$ system were systematically evaluated through cyclic experiments, mineralization analysis, and degradation tests of multiple pollutants. As shown in Figure S11, the catalytic performance gradually declined with repeated use, with BPA removal efficiencies decreasing from 98.6% in the first cycle to 86.3%, 80.7%, 75.0% and 64.3% in subsequent cycles. This attenuation can be primarily attributed to the partial blockage of active sites by accumulated intermediates and the consumption of surface reactive centers during continuous PI activation. In addition, the $\text{MnFe}_2\text{O}_4@\text{SBC}/\text{PI}$ system achieved a TOC removal efficiency of 69.2%, indicating its capability for mineralization of BPA (Figure S12).

To further assess the universality of this system, a range of organic contaminants were selected, including BPA, sulfamethoxazole (SCP), acetaminophen (ACE), and ciprofloxacin (CIP). The removal efficiencies were 98.6% (BPA), 98.2% (SCP), 95.9% (ACE), and 81.7% (CIP), respectively (Figure S13). The relatively lower degradation efficiency of CIP can be attributed to its more stable molecular structure and higher ionization potential, which makes electron abstraction and oxidative transformation less favorable compared with other pollutants. These results confirm that the $\text{MnFe}_2\text{O}_4@\text{SBC}/\text{PI}$ system exhibits broad-spectrum applicability toward diverse organic contaminants.

Moreover, the formation of toxicity of degradation products, iodine-containing by-products and the influence of coexisting inorganic anions were carefully examined and are presented in Text S9–S11. The results demonstrated that PI was predominantly converted into environmentally benign IO_3^- without generating toxic iodinated species. The presence of common anions exerted negligible effects on BPA degradation efficiency, indicating the environmental safety and robustness of the system.

4. Conclusions

In this study, a series of defect-engineered $\text{MnFe}_2\text{O}_4@\text{SBC}$ catalysts were successfully synthesized by coupling spinel MnFe_2O_4 with sewage sludge-derived biochar prepared at different calcination temperatures. Among them, $\text{MnFe}_2\text{O}_4@\text{SBC}$ -750 exhibited the best catalytic performance, achieving 98.6% BPA removal within 30 min during PI activation. Structural characterization revealed that the superior activity of $\text{MnFe}_2\text{O}_4@\text{SBC}$ -750 origi-

nated from its optimized carbon framework containing an appropriate balance between defect density and graphitization degree. This structural feature promoted interfacial electronic interaction between MnFe_2O_4 and SBC, thereby facilitating PI adsorption, electron transfer, and I-O bond activation. Mechanistic investigations demonstrated that $\bullet\text{OH}$, $\text{O}_2^{\bullet-}$, $\text{IO}_3^{\bullet}$ and $^1\text{O}_2$ jointly contributed to BPA degradation, while electrochemical analysis and DFT calculations further confirmed that defect-regulated π -d electronic coupling accelerated interfacial charge transfer and enhanced PI activation efficiency. Importantly, this work demonstrates that catalytic activity is not governed solely by defect abundance, but rather by the synergistic balance between structural defects and electronic delocalization. Overall, this study provides new insight into the role of defect regulation in spinel-carbon catalysts and offers a promising strategy for designing efficient biochar-based catalysts for advanced oxidation processes and environmental remediation.

Supplementary Materials: The following supporting information can be downloaded at: <https://www.mdpi.com/article/10.3390/w18111262/s1>, Figure S1: SEM images and EDS mappings of (a) MnFe_2O_4 , (b) SBC and (c) $\text{MnFe}_2\text{O}_4@\text{SBC}$; Figure S2: N_2 adsorption-desorption isotherms (BET analysis) and corresponding pore size distribution curves (BJH method) of (a) SBC, (b) MnFe_2O_4 and (c) $\text{MnFe}_2\text{O}_4@\text{SBC}$; Figure S3: Pseudo-first-order rate constants for BPA degradation under different conditions, including (a) $\text{MnFe}_2\text{O}_4@\text{SBC}$ synthesized with different MnFe_2O_4 : SBC ratios, (b) $\text{MnFe}_2\text{O}_4@\text{SBC}$ catalysts prepared from SBC calcined at different temperatures and (c) different systems; Figure S4: Pseudo-first-order rate constants for BPA degradation in the presence of different quenchers, including (a) MeOH, (b) p-BQ, (c) FFA, (d) phenol, (e) DMSO and (f) $\text{K}_2\text{Cr}_2\text{O}_7$; Figure S5: PI consumption in the presence of FFA; Figure S6: 2,4,6-TCP degradation efficiency in the $\text{MnFe}_2\text{O}_4@\text{SBC}/\text{PI}$ system (reaction conditions: $[\text{MnFe}_2\text{O}_4@\text{SBC}] = 0.4 \text{ g L}^{-1}$, $[\text{2,4,6-TCP}] = 20 \text{ mg L}^{-1}$, $[\text{PI}] = 0.5 \text{ mM}$, initial pH = 7; Figure S7: BPA removal efficiency under different pre-mixing times in the $\text{MnFe}_2\text{O}_4@\text{SBC}/\text{PI}$ system; Figure S8: High-resolution XPS spectra of $\text{MnFe}_2\text{O}_4@\text{SBC}$ before and after reaction: (a) C 1s, (b) O 1s, (c) Mn 2p and (d) Fe 2p spectra; Figure S9: Detailed symbol and atomic number of the BPA molecule; Figure S10: LC-MS spectra of intermediate products formed during BPA degradation in the $\text{MnFe}_2\text{O}_4@\text{SBC}/\text{PI}$ system at different reaction times; Figure S11: Reusability of $\text{MnFe}_2\text{O}_4@\text{SBC}$ for BPA degradation in five cycles; Figure S12: TOC removal efficiency of BPA in the $\text{MnFe}_2\text{O}_4@\text{SBC}/\text{PI}$ system; Figure S13: Degradation performance of various organic pollutants in the $\text{MnFe}_2\text{O}_4@\text{SBC}/\text{PI}$ system; Figure S14: (a) Full scan spectrum of SBC, high-resolution XPS spectra of (b) C1s, (c) O1s (d) Mn2p and (e) Fe2p of SBC; Figure S15: Effect of (a) catalyst dosage, (b) BPA concentration, (c) PI concentration and (d) initial pH on BPA degradation (reaction conditions: $[\text{MnFe}_2\text{O}_4@\text{SBC}] = 0.4 \text{ g L}^{-1}$, $[\text{BPA}] = 20 \text{ mg L}^{-1}$, $[\text{PI}] = 0.5 \text{ mM}$ and initial pH = 7); Figure S16: Pseudo-first-order rate constants for BPA degradation under different conditions, including (a) catalyst dosage, (b) BPA concentration, (c) PI concentration and (d) initial pH (reaction conditions: $[\text{MnFe}_2\text{O}_4@\text{SBC}] = 0.4 \text{ g L}^{-1}$, $[\text{BPA}] = 20 \text{ mg L}^{-1}$, $[\text{PI}] = 0.5 \text{ mM}$ and initial pH = 7); Figure S17: (a) I-t curves and (b) LSC on $\text{MnFe}_2\text{O}_4@\text{SBC}$ electrodes; Figure S18: T.E.S.T toxicity analysis of various intermediates: (a) oral LD50 in rats, (b) bioaccumulation factor, (c) developmental toxicity and (d) mutagenic developmental toxicity; Figure S19: (a) Changes in iodine species concentrations during the $\text{MnFe}_2\text{O}_4@\text{SBC}/\text{PI}$ reaction, (b) BPA removal performance in the $\text{MnFe}_2\text{O}_4@\text{SBC}/\text{NaIO}_3$ and $\text{MnFe}_2\text{O}_4@\text{SBC}/\text{KI}$ systems, (c) HOI formation in the $\text{MnFe}_2\text{O}_4@\text{SBC}/\text{PI}$ system, (d) Identification of I_2 and I_3^- species in the $\text{MnFe}_2\text{O}_4@\text{SBC}/\text{PI}$ system; Figure S20: Influence of different inorganic anions and HA on BPA degradation in the $\text{MnFe}_2\text{O}_4@\text{SBC}/\text{PI}$ system: (a) Cl^- , (b) SO_4^{2-} , (c) HCO_3^- , (d) CO_3^{2-} , (e) HPO_4^{2-} and (f) HA (reaction conditions: $[\text{MnFe}_2\text{O}_4@\text{SBC}] = 0.4 \text{ g L}^{-1}$, $[\text{BPA}] = 20 \text{ mg L}^{-1}$, $[\text{PI}] = 0.5 \text{ mM}$ and initial pH = 7); Figure S21: Kinetics of BPA degradation in the presence of various inorganic anions and HA in the $\text{MnFe}_2\text{O}_4@\text{SBC}/\text{PI}$ system: (a) Cl^- , (b) SO_4^{2-} , (c) HCO_3^- , (d) CO_3^{2-} , (e) HPO_4^{2-} and (f) HA (reaction conditions: $[\text{MnFe}_2\text{O}_4@\text{SBC}] = 0.4 \text{ g L}^{-1}$, $[\text{BPA}] = 20 \text{ mg L}^{-1}$, $[\text{PI}] = 0.5 \text{ mM}$ and initial pH = 7); Table S1: HPLC methods for the determination of organic pollutants and iodine species; Table S2: Specific surface area and average pore volume of $\text{MnFe}_2\text{O}_4@\text{SBC}$; Table S3: The comparison

of efficiency towards pollutants degradation in different systems. References [68–78] are cited in the Supplementary Materials.

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